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Researc h Article Design and Synth esis of Conducti ng Polym er Bio-B ased Polyuretha ne Produced from Palm Kernel Oil Muhammad Abdurrahman Munir , 1 Khairiah Haji Badri , 2,3 Lee Yook Heng , 2 Ahlam Inayatullah , 4 Hamid Alkhair Badrul , 4 Emelda Emelda , 1 Eliza Dwinta , 1 Nurul Kusumawardani , 1 Ari Susiana Wulandari , 1 Veriani Aprilia , 5 and Rachmad Bagas Yahya Supriyono 1 1 Department of Pharmacy, Faculty of Health Science, Alma Ata University, Daerah Istimewa Yogyakarta 55183, Indonesia 2 Department of Chemical Sciences, Faculty of Science and Technology, Universiti Kebangsaan Malaysia, Bangi 43600, Malaysia 3 Polymer Research Center, Universiti Kebangsaan Malaysia, Bangi 43600, Malaysia 4 Faculty of Science and Technology, Universiti Sains Islam Malaysia, Nilai 71800, Malaysia 5 Department of Nutrition Science, Alma Ata School of Health Sciences, Alma Ata University, Daerah Istimewa Yogyakarta 55183, Indonesia Correspondence should be addressed to Muhammad Abdurrahman Munir; muhammadabdurrahman2220@gmail.com Received 11 June 2021; Revised 1 October 2021; Accepted 23 March 2022; Published 7 April 2022 Academic Editor: Joanna Rydz Copyright © 2022 Muhammad Abdurrahman Munir et al.

This is an open access article distributed under the Creative Commons Attribution License, which permits unrestricted use, distribution, and reproduction in any medium, provided the original work is properly cited. Polyurethane (PU) is a unique polymer that has versatile processing methods and mechanical properties upon the inclusion of selected additives. In this study, a freestanding bio-based polyurethane ? Im the screen-printed electrode (SPE) was prepared by the solution casting technique, using acetone as solvent.

It was a one-pot synthesis between major reactants, namely, palm kernel oil-based polyol and 4,4-methylene diisocyanate. The PU has strong adhesion on the SPE surface.

The synthesized bio- based polyurethane was characterized using thermogravimetry analysis, di ? erential scanning calorimetry, Fourier-transform infrared spectroscopy (FTIR), surface area analysis by ? eld emission scanning electron microscope, and cyclic voltammetry. Cyclic voltammetry was employed to study electrocatalytic properties of SPE-polyurethane towards oxidation of PU.

Remarkably, SPE-PU exhibited improved anodic peak current as compared to SPE itself using the di ? erential pulse voltammetry method. Furthermore, the formation of urethane linkages (-NHC(O) backbone) after polymerization was analyzed using FTIR and con ? rmed by the absence of peak at 2241 cm -1 attributed to the sp-hydridized carbons atoms of C = C bonds. The glass transition temperature of the polyurethane was detected at 78.1 ° C. 1. Introduction Conducti ng polyme rs (CPs) are polymers that can releas e a c u r r e n t [1].

T h e c o n d u c t i v i t y o f C P s w a s ? rst observed in polyacetyle ne, nevertheless owing to its instabilit y, and the invention of various CPs has been studied and reporte d such as polyaniline (PANI), poly( o -tolu idine) (PoT), polyth io- phene (PTH), poly ? uorene (PF), and polyu rethane (PU). Furthermore, natural CPs have low conductivity and are often semiconduct ive. Ther efore, it is imperative to improve their conductivity for electroc hemical sensor purpose s [2 –4]. The CPs can be produced from many organic materials and th ey have several advantag es, such as having an electrical curr ent, inexpensive materials, mas sive surface area, and small dimensions, and the product ion is straightfor ward.

Further more, according to these properties, many studies have been reported by researche rs to study and report the variety of CPs applications such as sensors, bioc hemical app lications, elect rochromic devices, and solar cells [1, 5]. There is scienti? c documentation on the use of conductive polymers in various studies such as polyaniline [6], polyp yr-role [7], and polyuretha ne [8–11]. Hind International Journal ofcience Volume 2022, Article ID 6815187, 13 pages https://doi.org/10.1155/2022/6815187 Polyurethane productions can be obtained by

using sev- eral materials as polyols such as petroleum, coal, and crude oils.

Nevertheless, these materials have become very rare to ? nd and the price is very expensive at the same time required a sophisticat ed system to produce it. The reasons such as price and time consum ing to produce polyols have been considered by many researche rs; furthermore, ? nding utilizing plants that can be used as altern ative polyols should be done immediate ly [12]. Thus, to avoid the use of petro- leum, coal, and crude oils as raw materials for a polyol, veg- etable oils become a better choice to produce polyol in orde r to obtain a biodegr adable polymer. Vegetable oils th at are generally used for polyu rethane synthesis are soybean oil, corn oil, sun ? ower seed oil, coconut oil, nuts oil, rape seed, o l i v e o i l, a n d p a l m o i l [ 1 2, 1 3 ]. It is very straightfor ward for veget able oils to rea ct with a speci ? c group to produce a PU such as epoxy, hydroxyl, car- boxyl, and acrylate owing to the existen ce of (-C=C-) in veg- etable oils. Thu s, it provides appealing pro ? ts to veget able oils comp ared to petroleum considering the toxicit y, price, a n d h a r m t o t h e e n v i r o n m e n t [ 1 4, 1 5 ].

P a I m o i I b e c o m e s the chosen in this study to produce PU owing to it being largely cultivat ed in South Asia particularly in Malaysia and Indonesia. It has several pro ? ts comp ared to other veg- etable oils such as the easiest materials obtained, the lowest cost of all the common vegetable oils, and recognize d as the plantation that has a low environmenta I impact and removing CO 2 f r o m t h e a t m o s p h e r e a s a n e t s e q u e s t e r [ 1 6 , 17]. The application of bio-base d polymer has appealed much attention until now.

Global environmental activists have forced researchers to discover another material produc- i n g p o l y m e r s [18]. P U s h a v e m a n y a d v a n t a g e s t h a t h a v e been used by many researchers, they are not merely vers atile materials but also have the durabilit y of metal and the ? exi- bility of rubber. Furthermore, they can be promoted to replace rubber, metals, and plastics in several aspects.

Sever al applications of PUs have been reported and studied such as textiles, automotive, building and construct ion applications, a n d b i o m e d i c a l a p p l i c a t i o n s [19,20]. Poly u r e t h a n e s a r e a l s o considered to be one of the most useful material s with many pro? ts, such as possessing low conductivity, low den sity, absorption capab ility, and dimensional stability. They are a great research subject due to their mechanical, physical, and chemical properties [21– 24]. PU structure contains the urethane group th at can be formed from the rea ction between isocyanat e groups (-NCO) and hydroxyl group (-OH).

Nevertheles s, several groups can be found in PU structure such as urea, esters, ethers, and several arom atic groups. Furthermore, PUs can be produced from di ? erent sources as long as they contain speci ? c materia ls (polyol and methylene diphenyl diisocya- nate (MDI)), making them very useful for speci ? c appli ca- tions. Thu s, according to the desired properties, PUs can be divided into several types such as wat erborne, ? exible, rigid, coating, binding, sealants, adhesive s, and elasto- mers [25]. PUs are lighter th an other materials such as metals, gold, and platinum.

The hardness of PU also relies on the number of t h e a r o m a t i c r i n g s in t h e p o l y m e r s t r u c t u r e [26, 27], majo rly contributed by th e isoc yanate derivatives. PUs have also a conjugate structure where electrons can move in the main chain that causes electricity produc ed even the current is low. The current of conjugat ed linear (p) can be elab o- rated by the gap between the valence band and th e conduc- tion band, or called high energy level containin g elect rons (HOMO) and lowest ener gy level not containing elect rons (LUMO), r e s p e c t i v e l y [28, 29]. In the recent past, several conventional methods have been developed such as capillary elect rophoresis, liquid, and gas chromatogr aphy coupled with several detect ors.

Ne vertheless, although chromatogra phic and spectrometric app roaches are well developed for qualitative and quantita- tive analyses of analytes, several limitations emerged such as comp licated instrumentation, expensive, tedious sam ple prepa rations, and requiring large amounts of expensive sol- v e n t s t h a t w i l l h a r m t h e u s e r s a n d e n v i r o n m e n t [ 2 2 –24, 30 –33]. Ther efore, it is imperative to obtain and develop an alternative materia l that can be used to anal yze a speci? c ana lyte.

Electrochemica I meth ods are extremely promis ing meth ods in the determinat ion of an analyte in samples owing to the high selectiv ities, sensitivit ies, inexpensive, requir ements of small amounts of solvents, and can be oper- ate d by people who have no background in analytical chem- istry. In addition, sample preparation such as separatio n and extra ction step s are not needed owing to the selectivity of th is instrument wher e no obvious interference on the cur- rent response of seven seven

F e w w o r k s h a v e b e e n repo rted on the elect rochemical metho ds for th e determina- tion of analyte using electrodes combined with several elec- trode modi? ers such as carbon nanotube, gold, and graph ene [34, 35]. Nevertheles s, the materials are expensive and th e product ion is di ? cult. Thus, an electrochemic al app roach using inexpensive and eas ily available materia Is as electrode modi ? e r s s h o u I d b e d e v e I o p e d [ 3 6 , 3 7 ] . Nowadays, screen-printed elect rodes (SPEs) modi ? ed with condu cting polyme r have been developed for various elect rochemical sensing.

SPE becom es the bes t solution owing to the elect rode having several advantages such as fru- gal manu facture, tiny size, being able to produce on a large s c a l e, a n d c a n b e a p p l i e d f o r o n - s i t e d e t e c t i o n [ 3 8 ] . C o n - ductin g polyme rs (CPs) become an alternative to modifying th e screen-printed electrode s due to several advantages such as their electrical conductivity, able to captur e analyte by chemi cal/physical adsorption, and large surface area. Thus, CP s are very app ealing materials fro m electrochemic al per- spective s [39]. Such advantages of SPE encourage us to con- struct a new electrode for electrochemic al sensing, and no resea rch reporte d on th e direct electrochemica I oxidation of histamine using a screen-printed electrode modi ? ed by bio-b ased polyu rethane. Ther efore, this research is the ? rst to devel op a new electrode using (screen print ed polyu re- th ane electrode) SPPE without any condu cting materials. The purpose of th is work was to synthesize, characterize, and study the electrobeha vior of polyu rethane using cyclic voltam metry (CV) and di ? erential pulse voltam metry (DPV) attached to the screen-p rinted electrode.

To the best of our knowled ge, this is the e? rst attempt to use a modi? ed 2 Intern ational Journal of Polymer Science polyuretha ne electrode. The electroc hemistry of polyure- thane moun ted onto SPE is discus sed in detail. PU s are pos- sible to becom e an advanced fro ntier material that has been chemically modi? ed th e speci? c electrodes for bio/chemica I sensing appli cation. 2. Experimental 2.1. Chem icals 2.1.1. Synthe sis of Bio-Based Polyurethan e Film. Palm kernel oil- (PKO-p-) based polyol was supplied by the UKM Tech- nology Sdn Bhd through MPOB/UKM station plant, Pekan B a n g i L a m a, S e I a n g o r, a n d p r e p a r e d u s i n g B a d r i e t a I. [40] method.

4,4-diphenylmethan e diisocyanat e (MDI) was acquired from Cosmopo lyurethane (M) Sdn. Bhd., Klang, Malaysia. Sol vents and analytical rea gents were benzene ( = 99.8%), tolue ne ( = 99.8%), hexane ( = 99%), acetone ( = 99%), dimet hylsulfoxide (DMSO) ( = 99.9%), dimethylf or- mamide (DMF) ( = 99.8%), tetrahyd rofuran (THF) ( = 99.8%), and polyethyle ne glycol (PEG) with a molecular weight of 400 Da obtaine d from Sigma Aldrich Sdn Bhd, Shah Alam . 2.2. Apparatus.

Tensile testing was performed using a uni- versal testin g machine model Inst ron 5566 following ASTM D638 (Standar d Test Met hod for Tensile Properties of Plas- tics). The tensile propertie s of the polyuretha ne ? Im were measured at a velocity of 10 mm/min with a cell load of 5 kN. The thermal propertie s were performed using th ermo-gravimetry analysis (TGA) and di ? erential scanning calo- rimetry (DSC) analysis. TGA was performed using a thermal analyzer of the Perkin Elmer Pyris model with a heating rate of 10 ° C/mi n at a temperature range of 30 to 800 ° C under a nitrogen gas atmosph ere.

The DSC anal ysis was perform ed using a thermal analyzer of the Perkin Elmer Pyris mode I with a heating rate of 10 ° C/minute at a temper- ature range of -1 00 to 200 ° C under a nitrogen gas atmo- sph ere. Approximat ely, 5 –10 mg of PU was weighed. The sampl e was heated from 25 to 150 ° C for one minute, then coo led immediately from 150 to 100 ° C for another one minut e and ? nally, reheated to 200 ° C at a rate of 10 °

C/ min. At this point, the polyurethane encounters chan ging from elastic properties to brittle due to changes in the move-ment of the polymer chains.

Ther efore, the temperature in th e middle of the inclined regions is taken as th e glass tran-sition temperature (Tg). The melting tempera ture (Tm) is iden ti? ed as th e maximum endother mic peak by taking the are a below th e peak as the enthalpy point (?Hm). The morpholo gical analysis of PU? Im was performed by? eld emission scanni ng electron microscope (FESEM) mode I Gemini SEM microscope model 500-70-22.

Before the analysis was carried out, the polyuret hane ? Im was coa ted with a thin layer of gold to inc rease the conductivity of the ? Im. The coating meth od was carried out using a sputter- coater. The observations were conducted at a magni- ? cation of 200x and 5000x with 10.00 kV (electron high ten- sion (EHT)). The cro sslinking of PU was determ ined using the soxhlet extra ction metho d. About 0.60 g of PU sample was weighed and put in an extractor tube containing 250 mL of toluene , used as a solve nt. This ? ow of tolue ne was let runnin g for 24 h. Mass of the PU was weighed defore and after the re ? ux proc ess was carried out.

Then, the sample was dried in the conventio nal oven at 100 ° C for 24 h in order to get a con- stant mas s. The percentage of cro sslinking content known SPE modi?ed with polyurethane ?Im as working electrode Auxiliary electrode Reference electrode Figure 1: Potentiostat instrument to study the conductivity of SPE modi ? ed with polyurethane ? Im using voltammetric approach: CV and DPV. 3 International Journal of Polymer Science as th e gel content can be calculated using Equation n (1).

Gel content %  $\delta P = W \circ - W W \times 100 \%$ ,  $\delta 1 P$  where W o is the mass of PU before the re ? ux process (g) and W is the mass of PU aft er th e re ? ux process (g). FTIR spectroscopic analysis was perform ed using a Perkin-Elm er Spectrum BX instrument using the diamond attenuation total re ? ectance (DATR) method to con? rm the polyurethane , PKO-p, and MDI functional group. FTIR spectrosco pic analysis was performed at a wavenumb er of 4000 to 600 cm -1 to identify the peaks of the major func- tional groups in the formation of the polymer such as amide group (-NH ), urethane car bonyl group (-C =O), isocyanate group (-O=C=N -), and carbamate group (-CN). 2.3. Synthe sis of Polyurethan e. Firstly, the polyol prepol ymer solution was produced by combining palm kernel oil-b ased polyol and poly(ethy lene glycol) (PEG) 400 (100 : 40 g/g), and acetone 30% was used as a solution.

The compound was homoge nized using a centrifuge (100 rpm) for 5 min. Whereas diisocyanat e prepol ymer was obtaine d by mixing 4,4 ' -diph enylmethane diisocyanate (100 g) to acetone 30%, afterward the mixture was mix ed using a centrifuge for 1 min

to obtain a homogenized solution. Afterwar d, diiso- cyanate solution (10 g) was poured into a conta iner that con- tains polyol prepolymer solution (10 g) slowly to avoid an exothermic reaction occurring. The mixture was mixed for 30 sec until a homogenized solution was acquired.

Lastly, the polyurethane solution was poured on the electrode sur- face by using the casting metho d and dried at ambient tem- perature for 12 h. 2.4. Modi ? cation of Elect rode. Voltammetric tests were per- formed using Metrohm Autolab Software (Figure 1) ana- lyzer using cyclic voltammetry method or known as amperometr ic mode and di ? eren tial pulse voltammetry. All electrochemica l experiments were carried out using screen-p rinted electrode (diameter 3 mm) modi ? ed using polyuretha ne ? Im as working electrode, platinum wire as the auxiliary electrode, and Ag/AgCl electrode as a reference electrode.

All exper iments were conducted at a temp erature of  $20 \pm 2$  ° C. The PU was cast onto the screen-printed electrode (SPE) and analyzed using a single voltam metric cycl e bet ween -1200 and +1500 mV (vs Ag/AgCl) of ten cycles at a sca n- ning rate of 100 mV/s in 5 mL of KCl in orde r to study the activity of SPE and polyuret hane ? Im. Approximately, 0.1, 0.3, and 0.5 mg of polyu rethane were dropped separately onto the SPE surf ace, and dried at room temp erature.

The modi ? ed palm- based polyuret hane electrodes were then rinsed with deionize d water to remove physically adsorbed impu rities and resid ues of unreac ted material on the elec- trode surface. All electrochemica I mat erials and calib ration mea surements were carried out in a 5 mL glass bea ker with a con ? guration of three electrodes inside it. Platinum wire and silver /silver chloride (Ag/AgCl) electrodes we re used as auxilia ry and reference electrodes, while a screen-printed elect rode that had been modi? ed with polyuretha ne was app lied as a working electrode. 3.

Results and Discussion The synthesis of PU ? Ims was carried out using a prepo lymeriza tion metho d which involves the format ion of ure- th ane polyme r at an early stage. The reaction took place bet ween diisocya nate (MDI) and palm kernel oil-based polyol. Table 1 presents the PKO-p propertie s used in this study. The structural chain was extended with th e aid of poly(- ethyle ne glycol) to form ? exible and elastic polyu rethane ? Im.

In order to produce the urethane prepolym er, the iso- cyanate group (-NCO) attacks with the hydroxyl group (-OH) of polyol (PKO-p) while the other hydro xyl group of the polyol is attacked by the other isocya nate group [41] as shown in Figure 2. 3.1. FTIR Analysi s. Figure 3 shows the FTIR spectra for poly- uret hane, exhibiting the important

functional group peaks. A c c o r d i n g t o a <mark>s t u d y</mark> r e <mark>s e a r c h</mark> e d b y [ 4 1 ] , P K O - p <mark>r e a c t</mark> s w i t h MDI to form urethane prepolym ers. The NCO group on MDI reacts with the OH group on polyol whether PKO-p or PEG. It can be seen there are no important peaks of MDI in the FTIR spectr a.

This is furt her veri? ed by the absen ce of an absorption bands at the 2400 cm -1 belonging to MDI (-NCO groups). This could also con ? rm that the -N CO group on MDI had comp letely reacted with PKO-p to form the urethane -NHC(O) backbone. The presence of amide s (-NH), carbonyl uret hane group (-C=O), carbamate group (C-NH), and -C-O-C con? rmed the formation of ure- th ane chains. In this study, the peak of carbonyl urethane (-C =O) detect ed at 1727 cm -1 indic ated that th e carbonyl uret hane group was bonded without hydro gen owing to the hydrogen rea cts with the carbonyl urethan e group.

The reaction of polyurethane has been studied by Hamu- z a n a n d B a d r i [42] w h e r e t h e u r e t h a n e c a r b o n y l g r o u p w a s detect ed at 1730–1735 cm -1 while the MDI carbonyl was detect ed at 2400 cm -1. The absence of absorption bands at 2250–2270 cm -1 asso ciated with N=C=O bond str etching indic ates the absence of NCO groups. It shows that the poly-meriza tion reaction occurs entirely between NCO groups in M D I w i t h h y d r o x y l g r o u p s o n p o l y o l s a n d P E G [43].

T h e absen ce of peak s at 1690 cm -1 representing urea (C=O) in th is study indicated there is no urea for mation as a bypro- duct [44] of the polyme rization reaction that possib ly occurs due to the excess ive water. For the amin e (-NH) group, hydro gen bond to -NH and oxygen to form ether, and Table 1: The speci ? cation of PKO-p (Badri et al. [40]). Property Values Viscosity at 25 ° C (cps) 1313.3 Speci ? c gravity (g/mL) 1.114 Moisture content (%) 0.09 pH value 10–11 The hydroxyl number mg KOH/g 450 –470 4 Intern ational Journal of Polymer Science hydrogen bond to NH and oxygen to form carbonyl on ure- thane can be detect ed at the peak of 3301 cm -1 and in the wavenumb er at range 3326 –3428 cm -1 . This has also been studied and detected by Mutsuhisa et al. [45] and Lampman et . a l . [ 4 6 ] .

In thisrestein the search, the protonacceptor to nacceptor is carbon y l (-C=O) while the proton donor is an amin e (-NH) to form a hydrogen bond. The MDI chemical structure has the elec-trostatic capability that produces dipoles from several atom s such as hydrogen, oxygen, and nitrogen atoms. The charac-teristic of isocyanates causing them are highly reactive [47]. MDI was one of the isocyanate s used in this study, has an arom atic group, and is more reactive compared to ali-phatic group isoc yanates such as hexameth ylene diisocya-nate (HDI) or isophoro ne diisocyanate (IPDI).

Isocya nates have two groups of isocyanat es on each molecule. Diphenyl- methane diisocyanat e is an exception owing to its structure consisting of two, th ree, four, or more isocya nate groups [48]. The use of PEG400 in th is study as a chain nextender of the reference of polyure than is study as a chain optimal amount. The properties of polyure than are contributed by hard and soft copo lymer segments of both polyol monomers and MDI.

This makes the hard seg- ment of uretha ne serves as a crosslinking site bet ween the soft segment soft hepoly ol[47]. The mechanism of the prepolymerization in uret hane chains formation is a nucleophilic substitution reaction as studied by Yong et al. [49]. However, this study found amines as nucleophile s. Amine attacks carbon yl on isocya- nate in MDI in order to form two resonance structures of intermediate complexes A and B (Figure 4).

Intermediate complex B has a greater tendency to react with polyols due to stronger carbonyl (C=O) bonds than C=N bonds on interm ediate complexes A. Thus, intermediate complex B is more stab le than interm ediate complex A, as suggest ed b y p r e v i o u s r e s e a r c h e r s c o n d u c t e d b y W o n g a n d B a d r i [41]. Moreover, oxygen is more electronegative than nitrogen caus ing cations (H+) to tend to attack -CN bon ds compared to -CO.

The combina tion between long polymer chain and low crosslinking content gives the polymer elastic properties wher eas short-chain and high crosslinking produce hard and rigid polymers. Crosslinking in polyme rs consists of three- dimensio nal networks with high molecula r we ight. In some aspe cts, polyuret hane can be a macromolecu le, a giant mol- e c u l e [ 5 0 ] . However, complexes A and B intermediate were pro- duced after the nucleophile of PEG attacking the isocya nate group in the MDI.

However, PEG contains oxygen atoms that are more electron egative than nitrogen atoms inside the PKO-p chemical structure causing the reaction of nucle-ophilic substitution that occurs in PKO-p. Furthermore, amine has a higher probability of reacting compared to hydro xyl [51]. Amine with high alkalinity reacts with carbon at o m s o n M D I a s p r o p o s e d b y W o n g a n d B a d r i [41]. The production of interm ediate comp lexes unstabilizes the alkyl ions; nevertheless, the long carbon chains of PKO-p ensure the stability of alkyl ions.

The addition of PEG in this study is imperative, not merely to increase the chain leng th of PU but also to avoid the production of urea as a byproduct after the NCO group reacts with H 2 O from the environment. If the NCO group reacts with the excess wat er in the environment, the formation of urea and carbon dio xide gas will also occur

excessively (Figure 5). This reac- tion can cause a polyuret hane foam, not polyuret hane ? Im as we studied the ? Im. HO R1 OH + OCN R' NCO OCN R' N C H O O R1 O C N O R' NCO H + PEG 400 and Acetone CN O R' H N C H O O R1 O C N H R' N H C O diol O n Polyurethane O O Urethane prepolymer with isocyanate end group Diisocyanate + Accetone Polyol ? ? Figure 2: PU production via the prepolymerization method [41].

5 International Journal of Polymer Science 86 88 90 92 94 96 98 100 102 2950 3150 3350 3550 (%) T cm–1 \_e absorption bands of -NH (a) 45 55 65 75 85 95 105 2800 2850 2900 2950 3000 \_e peaks of alkene groups cm–1 (%) T (b) 0 20 40 60 80 100 120 1750 1950 2150 2350 \_e absorption bands of -NCO groups (%) T cm–1 (c) 60 70 80 90 100 110 1635 1685 1735 1785 \_e peaks of -C = O cm–1 (%) T (d) 63 68 73 78 83 88 93 98 1280 1300 1320 1340 \_e peaks of -C-N-H cm–1 (%) T (e) 45 55 65 75 85 95 105 900 1000 1100 \_e peaks of -C-O-O (%) T cm–1 PU MDI PKOp (f ) Figure 3: FTIR spectra of several important peaks between polyurethane, PKO-p, and MDI.

6 Intern ational Journal of Polymer Science Furthermore, th e app lication of PEG can in ? uen ce the c o n d u c t i v i t y o f P U w h e r e b y P o r c a r e I I i e t a I . [52] h a v e reported the application of PEG using several molecular weights. PEG 1500 decreased the condu ctivity of PU in con-sequence of the semicrystalline phase of PEG 1500 that acted as a poor ion-conduct ing phase for PU. It is also well known that PEG with a molecular weight of more than 1000 g · mol -1 tends to crystallize with deleterious e ? ects on room temper- a t u r e i o n i c c o n d u c t i v i t y [52]. 3.2. Morp hological Analysis.

The ? eld emission scanning electron microscope micrograph in Figur e 6 shows the formation of a unifo rm polyme r ? Im contrib uted by the poly- merization method appli ed. The magni ? cation used for this surface analysis rang ed from 200 to 5000x. The poly- merization method can also avoid the failure of the reaction in PU polyme rization. Furthermore, no trace of separation was detected by FESEM. This has also been justi ? ed by the wavelength s obtained by the FTIR spectra above. 3.3. The Crosslinking Analysi s. Soxhlet ana lysis was applied to determine the degree of cro sslinking between the hard segments and the soft segments in the polyuret hane.

The urethane group on the hard segment along the polyure- t h a n e c h a i n i s p o l a r [53]. The refore, d u r i n g t h e t e s t i n g, i t was very di? cult to dissolve in tolue ne, as the testing reagent. The degree of crosslinking is determined by the percentage of the gel content. The ana lysis result obtained from the soxhlet testing indica ted a 99.3% gel content. This is signi? cant in getting a stable polymer at a higher w o r k i n g t e m p e r a t u r e [54, 55]. Gel content %  $\delta P = 0: 6 - 0: 301 \delta P g 0: 301g \times 100\% = 99: 33\%: \delta 2 P 3.4$ . The Ther mal Analysis.

Thermogravimetr ic analysis can be used to observe the material mass based on temp era- ture shift. It can also examine and estimate the th ermal stab ility and materials propertie s such as the alteratio n we ight owing to abso rption or deso rption, decompos ition, reduction, and oxida tion. The material compositio n of polyme r is speci ? ed by analyzing th e temp eratures and t h e h e i g h t s o f t h e i n d i v i d u a l m a s s s t e p s [ 5 6 ].

F i g u r e 7 shows the TGA and derivat ive thermogravim etry (DTG) th ermograms of polyuret hane. The percenta ge weight loss (%) is listed in Table 2. Generally, only a small amount of weight was observe d. It is shown in Figure 7 in the regio n of 45 –180 ° C. This is due to the pres ence of conden- satio n on moistu re and solvent resid ues. T max : the temperature of polyuret hane started to degrade; T d 1 : thermal degra dation ? rst; T d 2 : thermal degra- dation second; T d 3 : thermal degradati on third.

The bio-base d polyurethane is thermally stab le up to 240 ° C b e f o r e i t h a s u n d e r g o n e t h e r m a l d e g r a d a t i o n [57]. The ? rst stage of thermal degrada tion (T d 1) on polyurethane ? Ims was shown in the region of 200 –290 ° C as shown in Figur e 7. The T d 1 is associated with degradation of the hard segm ents of the uretha ne bond, forming alcohol or degrada-tion of t h e poly olchains and releas i n g o f i s o c y a n a t e s [58], primar y and secondary amines, as well as carbon dioxide [59, 60]. M e a n w h i l e, the second the r m a l d e g r a d a t i o n s t a g e (T d 2) of polyurethane ? Ims experien ced a weight loss of 39.29%.

This endother m of T d 2 is relate d to the dimerizat ion of isocya nates to for m carbodiimide s and release CO 2. The for med carbodiimide reacts with alcohol to form urea. The H2 C NCO N C O NH C HOH2CH2C CnH2n+1n O HOH2CH2C H2 C NCO N C O NH C HOH2CH2C CnH2n+1n O HOH2CH2C Intermediate complex A Intermediate complex B + - + Figure 4: The formation of intermediate complexes. NCO + H2O Step 1 R NH C OH Step 2 Degradation R O R NH2 + CO2 Figure 5: The reaction between the NCO group and water producing carbon dioxide. Figure 6: The micrograph of polyurethane ? Ims was analyzed by FESEM at (a) 200x and (b) 5000x magni ? cations.

7 International Journal of Polymer Science third stage of thermal degrada tion (T d 3) is related to the deg- radation of urea [58] and the soft segment on polyurethane. Generally, DSC analysis exhibited thermal transitions as well as the initial crystallization and me lting temp eratures of the poly ure than e [61]. It servest o an aly z e c h a n g e s in t h e r - mal behavior due to changes occurring in the chemical chain structure based on the T g of the sample obtained from the DSC thermogram (Figure 8).

DSC analysis on polyuret hane ? Im was perform ed in the temperature at the range 100 ° C to 200 ° C of using nitrogen gas as a blanke t as proposed by F u r t w e n g l e r e t a I. [62]. T h e g l a s s t r a n s i t i o n t e m p e r a t u r e o n polyuretha ne was above room temperature , at 78.1 ° C indi- cated the state of glass on polyuretha ne. The presence of MDI contr ibutes to the formation of hard segm ents in poly- u r e t h a n e s. P o r c a r e l l i e t a l. [52] s t a t e d t h a t p o s s e s s i n g a l o w T g may contrib ute to PU conductivity.

During polymeriza tion, this hard segment restricts the m o b i l i t y o f t h e p o l y m e r c h a i n [ 6 3 ] o w i n g t o t h e s t e r i c e ? ec t on the benz ene ring in the hard segment. The endother mic peak of acetone used as the solve nt in this study was suppos- edly at 56 ° C. However, the acetone peak was not detected in the DSC and TGA instruments, which can be concluded the acetone was completely evaporated duri ng the polyuret hane synthesis. The presence of acetone in the synt hesis was to lower the rea ction kinetics.

3.5. The Solubi lity and Mechanical Properties of the Polyurethane Film. The chemi cal resistivity of a polyme r will be the determinant in performing as a conductor. Thus, its solubility in various solve nts was determined by dissolving the polymer in selected solvents such as hexane, benzen e, acetone, THF, DMF, and DMSO. On the other hand, the mechan ical properties of polyu rethane were determined based on the standard testing followin g AS TM D638. The resu lts from the polyuret hane ? Im solubility and tensile test are shown in Tabl e 3.

Polyurethan e ? Ims were insoluble with acetone , hex ane, and benzene and are only slightly soluble in THF, DMF, and DM SO solutions. While the tensile strength of a PU ? Im indic ated how much elongation load the ? Im was capable of withstanding the material before breaking. The tensile str ess, strain, and modulus of polyurethane ? Im also indica ted that polyurethane has good mechanical properties that are capable of being a supporting subst rate for the next stage of the study.

In the product ion of polyure- th ane, the propertie s of polyurethane are eas ily in ? uenced by th e content of MDI and polyol used. The length of the chain and its ? exibility are contrib uted by the polyol which makes it elastic. Hig h crosslinking content can also produce hard and rigid polymers. MDI is a major component in th e for- matio n of hard segm ents in polyurethane . It is this hard seg- ment that determines the rigidity of the PU. Ther efore, high isoc yanate content results in higher rigidity on PU (Petr ovic et al. 2002). Thus, the polyme r has a higher resistance to deform ation and more stress can be applied to the PU. 3.6. The Conductivit y of the Polyurethane as a Polymeric Film on SPE. Polyurethan e ? Im was depos ited onto the screen-p rinted elect rode by casting method as shown in Figur e 1. After that, the modi? ed electrode was analyzed using cyclic voltam metry and di ? erential pulse voltam metry in order to study the behavior of the modi? ed electrode. The modi ? ed electrode was tested in a 0.1 mmol  $\cdot$  L -1 KCl solut ion containin g 5 mmol  $\cdot$  L -1 (K 3 Fe(CN) 6 ). The use of potas sium ferric yanide is intended to increase the sensitivit y of the KCl solution. The conductivity of the modi ? ed electrode was studied.

The electrode was anal yzed by cyclic voltamm- etry method with a potential range of -1.00 to +1.00 with a scan rate of 0.05 V·s -1. The voltammograms at the electrode have shown a speci? c redox reaction. Furthermor e, the con-ductivity of the modi? ed electrode is lower due to the use of polyu rethane. This occurs due to PU being a natural poly-mer produced from the polyol of palm kernel oil-based -5 -4 -3 -2 -1 0 1 0 100 200 300 400 500 600 700 800 900 Weight (%) Temperature (°C) Figure 7: DTG thermogram of polyurethane? Im.

Table 2: Weight loss percentage (wt%) and thermal degradation (Td) of polyurethane ? Im. % weight loss ( wt %) and thermal degradation (Td) Total of weight loss (%) Residue after 550 ° C (%) T max , ° C ðÞ T d 1 , 200 – 290 ° C T d 2 , 350 – 500 ° C T d 3 , 500 – 550 ° C 240 8.04 39.29 34.37 81.7 18.3 8 Intern ational Journal of Polymer Science polyol. The electrochemica I signal at the electrode is low if there is a decrea se in electrochemica I conductivity [64]. It can be concluded that polyurethane is a biopolymer with a low current value.

The current of th e modi ? ed elect rode was found at 5 :  $3 \times 10 - 5$  A or  $53 \mu$  A. Nevertheles s, the cur- rent of PU in this study showed better results comp ared to Bahrami et al. (2019) that reported the current of PU as  $1 : 26 \times 10 - 6$  A, where as L i e t a l . [ 6 5 ] r e p o r t e d t h e P U c u r r e n t in their study was even very low, namely, 10 -14 A. The PU can obtain a current owing to the benz ene ring in the hard segment (MDI) could exhibit the current by inducing elec- tron delocalization along the polyuret hane chain [66]. The PU can also releas e a current caused by PEG.

The applica- tion of PEG as polyol has been studied by Porcarelli et al. [52] t h a t r e p o r t e d t h a t t h e c u r r e n t o f P U b a s e d o n P E G – polyol was  $9: 2 \times 10 - 8$  A. According to Figure 9, it can be concluded that the anodic peak present in the modi? ed elect rode was at +0.5 V, and it also represented the anodic peak of the SPE- PU. The ? rst oxidation signal on both electrodes ranged from -0.2 to +1.0 V, which revealed a particu lar oxidative peak at a potential of +0.5 V. Figure 10 also pres ents th e DPV voltam mogram of the modi ? ed electrode. DPV is a measurement based on the differenc e in potentia l pulses that produce an electric current.

Scan ning the capability pulses to the working electrode will produce di ? erent currents. Optimal peak curr ents will be produced to the reduction capacity of the redox material . The peak current produced is proporti onal to the concentra- tion of the redox substance and can be detected up to a con- ce ntration below 10 -8 M. DPV was conducted to obtain the current valuet hat is more a curate than CV [ 67]. This study used a redox pair (K 3 Fe(CN) 6) as a test device (probe).

The currents generated by SPE-P U and prove d by CV and DPV have shown conductivity on polyu rethane ? Ims. This suggests that polyu rethane ? Ims can conduct elect ron transfer. The electrochemica I are a on the modi ? ed elect rode can be calculat ed using the formula from R a n d I e s - S e v c i k [ 6 8 ] , w h e r e t h e e I e c t r o c h e m i c a I a r e a f o r S P E - P U i s c o n s i d e r e d t o b e A , u s i n g E q u a t i o n ( 3 ) : C ur r ent of SP E - PU , Ip =2 : 65 × 105 AC n 3/2 v 1/2 D 1/2 , ð3 P where n – 1 is the amount of electron transfer involve d, while C is the solve nt concentratio n used (mmol · L -1 ) and th e valu e of D is the di ? usion constant of 5 mmol · L -1 at (K 3 Fe(CN) 6 ) dissolved using 0.1 mmol · L -1 KCI. The esti- mated surface area of the elect rode (Figure 1) was 0.2

cm 2, wher e the length and width of the electrode used duri ng th e study was 0 : 44 cm × 0 : 44 cm , while the surface area of th e SPE-P U was 0.25 cm 2 with the length and width of the elect rode estimat ed at 0 : 5 cm × 0 : 5 cm , causing the SPE-P U has a large r surface. The corresponding surface concentra- tion (t) (m ol/cm 2) is m e a s u r e d u s i n g E q u a t i o n (4). Ip = n 2 F 2 4 RT \_\_ A t v , ð4 P where I p is the peak curr ent (A), while A is the surface are a of the electrode (cm 2), the value of v is the applied scan rate (mV/s) and F is the Faraday constant (96,584 C/mol), R is the const ant ideal gas (8.314 J/mol K), and T is the temper- ature used during the exper iment being condu cted (298 K) -4.5 -4 -3.5 -3 -2.5 -2 -1.5 -1 -0.5 0 0.5 20 40 60 80 100 120 140 160 T (mW) Temperature (°C) Onset Figure 8: DSC thermogram of polyurethane ? Im.

Table 3: The solubility and mechanical properties of the polyurethane ? Im. Parameters Polyurethane ? Im Solubility Benzene Insoluble Hexane Insoluble Acetone Insoluble THF Less soluble DMF Less soluble DMSO Less soluble Stress (MPa) 8.53 Elongation percentage (%) 43.34 Strain modulus (100) (MPa) 222.10 9 International Journal of Polymer Science [ 6 9 ] . T h e a p p l i c a t i o n o f P K O - p t o p r o d u c e a c o n d u c t i n g polymer will be a great prospect as this material can be employed in the anal ytical indust ry in order to modify elec- trodes for electrochemic al purposes. Furthermore, a number of palm oils are abundant in Malaysia and Indonesia such as palm stearin and re ? ned- bleached- deodorized (RBD) palm oil. They have several bene ? ts such as being sustainable, cheap, and environmental ly biodegradable . These palms are th e pote ntial to produce bio- materials that can be used to repla ce other polymers that are chemical based [17]. Several studie s have been reported the application of PU to produce elastic condu ctive ? bres and ? Ims owing to it being highly elastic , scratch -resistant, and a d h e s i v e [70]; t h u s, i t i s e a s y f o r P U t o a d h e r e t o t h e screen-p rinted electrode to modify the electrode. PU is also being used as a composite material to mak e elastic con- ducting composite ? Ims [71]. 4.

Conclusion Polyurethane ? Im was prepared by prepolym erization between palm kernel oil-b ased polyol (PKO -p) with MDI. The presence of PEG 400 as the chain extende r formed free- standing ? exible ? Im. Acetone was used as the solvent to lower the reaction kinetics since the prepolymerization was car ried out at room temp erature. The formation of urethane link s (-NHC(O) backbone) after polyme rization was con- ? rmed by the absence of absorption bands at 2241 cm -1 asso- ciat ed with the N=C= O bond stretching and the pres ence of N- H peak at 3300 cm -1, carbonyl (C=O) at 1710 cm -1, carba- mate (C-N) at 1600 cm -1, ether (C-O-C) at 1065 cm -1, ben- ze ne ring (C=C) at 1535 cm -1 in the bio-based polyu rethane chain str ucture.

So xhlet analysis for the deter- mina tion of cro sslinking on polyurethane ? Ims has yielded a high percentage of 99.33% . This is contributed by the hard segm ents formed from the reaction between isocyanates and hydroxyl groups causing elongation of polymer chains. FESE M analysis exh ibited an absence of phase separatio n and smooth surface. Meanw hile, th e current of the modi ? ed elect rode was found at  $5:2 \times 10 - 5$  A . This bio-base d polyure- th ane ? Im can be used as a condu cting biopolymer, and it is very useful for other studie s such as electrochemic al sensor purp oses.

Furthermor e, advanced tech nologies are promis - ing, and the future of bio-base d polyol looks very brigh t. Data Availability All data experiments can be found in the manuscript. -100 - 80 - 60 - 40 - 20 0 20 40 60 - 1.2 - 0.7 - 0.2 0.3 0.8 Current (  $\mu$ A) E (V) Figure 9: The voltammogram of SPE-PU modi ? ed electrode after analyzed using cyclic voltammetry technique. 0.40 0.90 1.40 1.90 2.40 2.90 3.40 0.00 0.20 0.40 0.60 0.80 Current (  $\mu$ A ) E ( V ) Figure 10: The voltammogram of SPE –PU modi? ed electrode after analyzed after analyzed using di ? erential pulse voltammetry technique. 10 Intern ational Journal of Polymer Science Conflicts of Interest The authors declare no con ? ict of interest.

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